

Water Oxidation by λ -MnO₂: Catalysis by the Cubical Mn₄O₄ Subcluster Obtained by Delithiation of Spinel LiMn₂O₄

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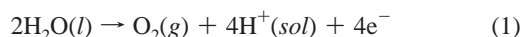
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Abstract: Nanocrystalline spinel LiMn₂O₄ has been prepared and treatment of LiMn₂O₄ with dilute nitric acid solution resulted in the delithiation of the framework, while maintaining the spinel structure, λ -MnO₂. LiMn₂O₄ is not a catalyst for water oxidation. Upon removal of the lithium, the cubical Mn₄O₄ cores become active sites for oxidizing water to molecular oxygen, which was investigated with the photochemical [Ru²⁺(2,2'-bpy)₃]/persulfate system at pH 5.8. The nanosize λ -MnO₂ obtained from the nanocrystalline LiMn₂O₄, which was synthesized by the citrate route, shows a significantly higher water oxidation catalytic activity (Turnover Frequency: 3×10^{-5} mol O₂/s/mol Mn) than that obtained via solid state reaction with micrometer and irregular particle sizes (Turnover Frequency: 5×10^{-6} mol O₂/s/mol Mn).

New catalysts made from inexpensive and abundant materials are needed to replace precious metals currently used in commercial water electrolyzers for the production of hydrogen and oxygen. RuO₂ and IrO₂ are the most widely used in commercial electrolyzers,¹ despite their limited availability and high cost. Many efforts have sought to understand and develop the potential of these materials.^{2–4}

Another approach taken thus far for the oxygen evolving half-cell reaction (OER, eq 1) is to model catalysts upon the photosystem II water-oxidizing complex (PSII-WOC), found within photosynthetic organisms.



The universally conserved catalytic core of this enzyme is comprised of a cubical CaMn₃O₄ cluster and a fourth Mn atom bridged via an oxygen atom.^{5,6} The chemical principles that may govern catalysis by this inorganic core have been studied through development of homogeneous water oxidation catalysts that are structurally related, including cubical Mn₄O₄ core molecules^{7,8} and less related Mn₂O₂ core molecules.^{9,10}

In addition, metal oxides of various compositions and allotropes have been studied for water oxidation potential, including spinels which exhibit low catalytic activity in the bulk phase.^{4,11} Recently spinel-type Co₃O₄ nanoparticles have demonstrated enhanced catalytic capabilities relative to the bulk phase.^{12,13} Although manganese oxides (Mn₂O₃ and rutile MnO₂) have shown little success in water oxidation trials,^{4,11,14} the concepts learned from the PSII-WOC and the Mn₄O₄ core complex encourage further studies of these materials.

Herein, we examine the possibility of activating the B site cations in AB₂O₄ spinels for catalysis by removal of the A site cations. Because the B cations are arrayed as B₄O₄ cubes resembling the PSII-WOC and the Mn₄O₄-cubanes, we are testing whether the cubical topology is the common feature for catalysis of water

oxidation. We report the first observation, to our knowledge, that removal of the Li cation from LiMn₂O₄ converts this inactive spinel to a water oxidation catalyst.

LiMn₂O₄ is a highly studied material used as a cathode in rechargeable batteries with many nanoscale synthesis procedures. It has a spinel type structure with Mn^{III} and Mn^{IV} ions occupying the octahedral B sites and Li ions in the tetrahedral A sites as seen in Figure 1.

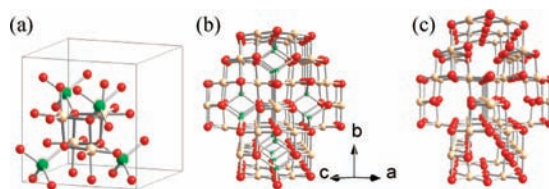
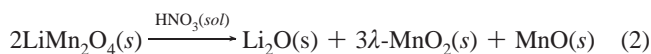


Figure 1. Perspective view of (a) a cubane core in the unit cell of spinel LiMn₂O₄, (b) an extended three-dimensional framework structure of LiMn₂O₄, and (c) λ -MnO₂ with void spaces after removal of Li ions. Li, Mn, and O atoms are shown as green, pink, and red, respectively.

The exchange properties of Li⁺ from LiMn₂O₄ are well understood.^{15–17} Li⁺ can be fully removed from the spinel framework yielding an allotrope of MnO₂ denoted λ -MnO₂. This material retains the spinel framework but with empty A sites, resulting in a uniquely open structure. λ -MnO₂ is not found naturally and differs from the common allotrope β -MnO₂ (rutile structure, with all O atoms tricoordinate). The B cations in λ -MnO₂ and LiMn₂O₄ are organized as cubical Mn₄O₄ subunits that are linked to the other B site cations via oxo bridges (exclusively dicoordinate in λ -MnO₂). The cubical Mn₄O₄ units in λ -MnO₂ are topologically similar to the Mn₄O₄ core found in the molecular “cubane” catalysts used for water oxidation and indirectly the CaMn₄O₄ core of the PSII-WOC structure.

We used two synthetic procedures to generate nanoscale LiMn₂O₄, which are described in the Supporting Information. Briefly, the first procedure utilizes a high temperature (850 °C, HT) route with Li₂CO₃ and Mn₂O₃.¹⁷ The second combines Mn(OAc)₂ with LiNO₃ at a lower temperature (350 °C, LT) in the presence of urea and citrate in acidic solution to aid in forming a higher surface area material during degassing of H₂O, NH₃, and CO₂.¹⁸ Removal of the Li⁺ is performed by dilute HNO₃ solution treatment¹⁷ as described below and in the Supporting Information. The acid treatment (eq 2) dissolves the Li₂O and MnO products of the reaction and yields a solid that analyzes gravimetrically as λ -MnO₂ with 100% removal of Li⁺.



Powder X-ray diffraction verifies the formation of spinel LiMn₂O₄ and reveals that the spinel spacing is maintained in

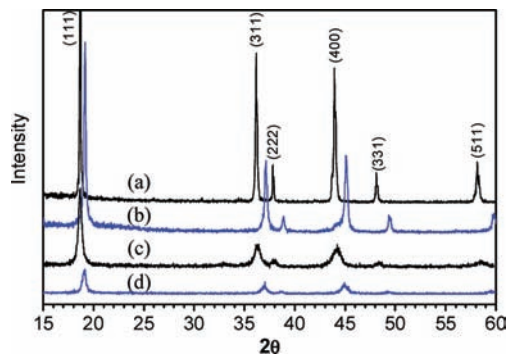


Figure 2. X-ray diffraction patterns of (a) LiMn_2O_4 (HT), (b) $\lambda\text{-MnO}_2$ (HT), (c) LiMn_2O_4 (LT), and (d) $\lambda\text{-MnO}_2$ (LT). Diffraction peaks of the $\lambda\text{-MnO}_2$ material show an equivalent upshift of 2θ .

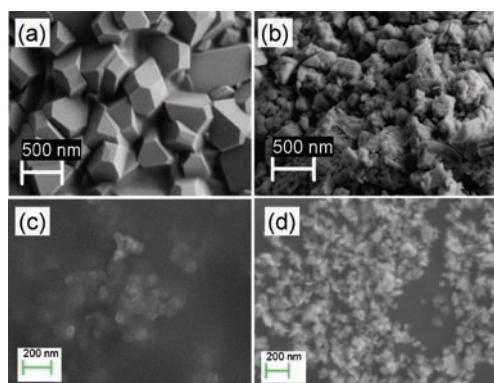


Figure 3. SEM images of (a) LiMn_2O_4 (HT), (b) $\lambda\text{-MnO}_2$ (HT), (c) LiMn_2O_4 (LT), and (d) $\lambda\text{-MnO}_2$ (LT).

$\lambda\text{-MnO}_2$ after delithiation as shown in Figure 2. All diffraction peaks of the $\lambda\text{-MnO}_2$ material show an equivalent upshift of 2θ equal to about 3% and broadening, while maintaining the spinel pattern as reported in the literature.¹⁷ The retention of the spinel pattern and complete delithiation upon acid treatment confirm the λ phase MnO_2 material. Upon removal of the A site atoms, the cubical structure highlighted in Figure 1a is maintained while creating nanochannels with an approximate dimension of $4 \text{ \AA} \times 3 \text{ \AA}$ (Figure 1c).

Scanning Electron Microscopy (SEM) of the product of the high temperature method reveals crystallite particle sizes ranging from 200 to 500 nm in diameter (Figure 3). The treatment of the high temperature material with HNO_3 produced a delithiated material with a fractured crystal structure and visible imperfections on the crystal faces. The low temperature synthesis produces a wide range of sizes and more irregular crystallites with the smallest feature at ~ 20 nm, as seen in the SEM and confirmed by the peak width in XRD. The crystallite size of the LiMn_2O_4 (LT) and $\lambda\text{-MnO}_2$ (LT) powders was calculated by Scherrer's formula with the full width at half-maximum data of the (111) plane observed at $18.68^\circ 2\theta$, yielding an average crystallite size of ~ 22 and ~ 19 nm, respectively.

Scheme 1. Photon Driven Oxidation System Used for Solution Phase Water Oxidation Experiments: One Electron per Turnover $\text{Ru}^{\text{III}}/\text{Ru}^{\text{II}}$

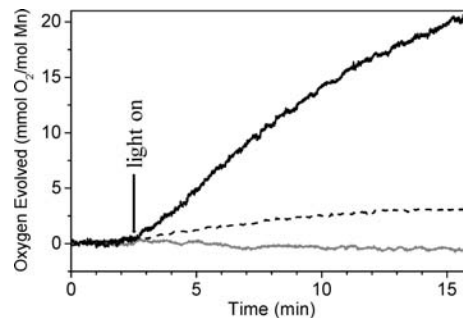
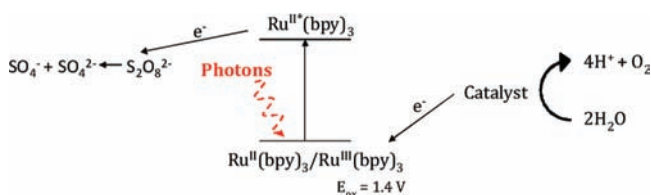


Figure 4. Oxygen evolution by Clark electrode measurements for $\lambda\text{-MnO}_2$ (LT; black), $\lambda\text{-MnO}_2$ (HT; dashed), and LiMn_2O_4 (gray).

Catalytic water oxidation was monitored in solution through detection of dissolved O_2 by thermostatted Clark-type electrode and confirmed by gas chromatography as described in the Supporting Information. Oxidative equivalents were provided upon illumination through a standard photoexcitation system dissolved in water¹² as depicted in Scheme 1.

Oxygen concentrations were recorded for over 15 min upon illumination as shown in Figure 4, from which initial rates were determined. Separate experiments were performed with reaction solutions without each of the key components of the photocatalytic system (catalyst, $\text{Ru}(\text{bpy})_3^{2+}$, light, persulfate) yielding no oxygen evolution. The requirement for a complete system for oxygen evolution indicates the reaction requires each of these components.

Figure 4 clearly shows that removal of Li from the LiMn_2O_4 structure creates a catalytically active $\lambda\text{-MnO}_2$ material from an otherwise inactive spinel. Neither the HT nor LT LiMn_2O_4 material exhibits activity in water oxidation. However, by decreasing particle size and morphology of the $\lambda\text{-MnO}_2$ material from 200–500 nm (HT) to ca. 20–100 nm (LT) (Figure 3) there is a 6-fold increase in the initial rate of O_2 evolution, suggesting that surface accessibility to the photocatalytic system limits the turnover rate. The one-dimensional channels produced within the $\lambda\text{-MnO}_2$ ($4 \text{ \AA} \times 3 \text{ \AA}$, Figure 1) are not large enough for permeation of the $\text{Ru}(\text{bpy})_3^{3+}$ oxidizing agent. Thus only the surface Mn species likely participate in the photoreaction, and removal of Li and particle size are the mechanistically important attributes for catalysis at the atomic level.

Table 1. Turnover Frequency (TOF) Numbers for Various Mn- and Co-Based Bulk Catalysts (mol O_2 /s/mol). In Parenthesis are Surface Mn or Co (mol O_2 /s/mol)

Catalyst Material	TOF (s^{-1})
$\lambda\text{-MnO}_2$ (HT)	5×10^{-6} (0.001) ^a
$\lambda\text{-MnO}_2$ (LT)	3×10^{-5} (?) ^a
SBA-15/ Co_3O_4 (4%) ¹²	NA (0.01)
Co_3O_4 nanocubes ¹³	0.09 (0.12) ^b
$\text{Mn}_2\text{O}_3(\text{terpy})_2(\text{OH})_2$ ⁹	5×10^{-5} (NA)
$\text{Mn}_4\text{O}_4(\text{MeOPh}_2\text{PO}_2)/\text{Nafion}$ ⁸	2.4×10^{-4} (NA)

^a The intrinsic TOF for $\lambda\text{-MnO}_2$ is higher, because in our method the rate is limited by the photogeneration of Ru^{III} . ^b All TOF values are at neutral pH except for Co_3O_4 nanocubes (ref 13), which is at pH 14 (5 M KOH) and an overpotential of 388 mV.

Initial oxygen evolution rates for the $\lambda\text{-MnO}_2$ (LT and HT) are given in Table 1. The values calculated for oxygen evolution rates correspond to low end estimates for turnover based on every Mn atom catalytically active. For the HT phase we estimate a TOF of $1 \times 10^{-3} \text{ s}^{-1}$ per surface Mn atom, based upon the surface/volume ratio of the crystallites (Supporting Information). The value for TOF per surface Mn for $\lambda\text{-MnO}_2$ (HT) is 10 times lower than that for the much smaller SBA-15/ Co_3O_4 spinel ($7.6 \text{ nm} \times 50 \text{ nm}$) (Table

1), which represents a dilute solid-state solution, not a concentrated solid like λ -MnO₂.¹² Comparisons to other well characterized catalytic water splitting systems (Table 1) reveal that λ -MnO₂ is an important step toward a robust Mn-based water-oxidizing catalyst that uses nontoxic, earth-abundant materials. We conclude that the cubical B₄O₄ unit in spinels can be activated by removal of structural Li⁺, which introduces flexibility as the important degree of freedom for catalysis. This design principle is analogous to that believed to operate in the PSII-WOC and molecular cubanes.⁸

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Supporting Information Available: Synthesis procedures for LiMn₂O₄ and λ -MnO₂. Experimental details for oxygen evolution procedure. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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